

Optical conductivity of rattling phonons in type-I clathrate $\text{Ba}_8\text{Ga}_{16}\text{Ge}_{30}$

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(Dated: May 22, 2009)

A series of infrared-active optical phonons have been detected in type-I clathrate $\text{Ba}_8\text{Ga}_{16}\text{Ge}_{30}$ by terahertz time-domain spectroscopy. The conductivity spectra with the lowest-lying peaks at 1.15 and 1.80 THz are identified with so-called rattling phonons, i.e., optical modes of the guest ion $\text{Ba}^{2+}(2)$ with T_{1u} symmetry in the oversized tetrakaidecahedral cage. The temperature dependence of the spectra from these modes are totally consistent with calculations based on a one-dimensional anharmonic potential model that, with decreasing temperature, the shape becomes asymmetrically sharp associated with a softening for the weight to shift to lower frequency. These temperature dependences are determined, without any interaction effects, by the Bose-factor for optical excitations of anharmonic phonons with the nonequally spaced energy levels.

PACS numbers: 63.20.Ry, 78.30.-j, 82.75.-z

During the past decade, thermoelectric materials such as clathrates and filled-skutterudites have renewed an interest in phonons[1]. These conductors are formed by polyhedral building blocks, where each polyhedral cage accommodates a guest ion like an alkali-earth or rare-earth element. If the cage is oversized, the guest ions can vibrate with large amplitude around an on-center or off-center site in the cage potential. These Einstein-like local modes have been called *rattling phonons*. The renewed interest above has been paid to interactions of these rattling phonons with acoustic phonons propagating through the cage network and carrying heat entropy, and, equally or more interestingly, with conduction electrons. However, these issues are still far from being well understood that even the charge dynamics of a single ion in rattling vibration have been little known.

This paper reports on the optical conductivity of rattling phonons detected in a type-I clathrate $\text{Ba}_8\text{Ga}_{16}\text{Ge}_{30}$ (hereafter abbreviated as BGG), featuring the anharmonicity effects on the conductivity spectra in comparison with model calculations.

This compound belongs to a family of ternary type-I clathrates having cubic crystal structure with space group $Pm\bar{3}n$ [2]. The unit cell of the host framework consists of 6 tetrakaidecahedra and 2 dodecahedra. The latter smaller cages around $2a$ sites occupy the *body-centered-cubic* sites, while the former oversized cages around $6d$ sites line up along the principal axes by sharing both the pentagonal faces with the dodecahedral cages and the hexagonal faces with neighboring tetrakaidecahedra. Every cage encapsulates one Ba^{2+} ion, and the guest ions in the smaller ($2a$) and oversized ($6d$) cages

are defined as $\text{Ba}^{2+}(1)$ and (2), respectively. These guest ions satisfy the Zintl rule for charge compensations that the more electro-positive guest atom donates electrons to the more electro-negative cage; 16 electrons are transferred from 8 Ba atoms to 16 Ga atoms of the cage. With finely tuned Ga/Ge concentration, therefore, the system can be a *heavily* doped semiconductor, in general, having both the charge-carrier's sign and density controlled.

So far the electrical resistivity, thermal conductivity and specific heat measurements have been made on BGG having different carrier densities of *n*- or *p*-type carriers[3, 4]. The nuclear density map for the $\text{Ba}^{2+}(2)$ ion in the oversized cage obtained by neutron diffraction[3] indicates that $\text{Ba}^{2+}(2)$ resides on center, while other ions, Sr and Eu, move away to off-center sites, being supported by the band structure calculations[5]. On the contrary, recent more detailed structural analyses[6] indicate that the $\text{Ba}^{2+}(2)$ ions are slightly displaced from the center depending on temperature and also carrier-type. This structural indication seems to be consistent with Raman scattering measurements of low lying additional modes[7].

Single crystals of *p*-type BGG are grown by a self-flux method[8]. To obtain transmitting signals as strong as possible through the conductive sample, single crystal disks of 2-5 mm in diameter and 0.5-1.0 mm in thickness are polished down to about 20 μm in thickness using diamond abrasive films. Terahertz time-domain spectroscopy (THz-TDS) measurements covering the frequency range of 0.2 - 3.5 THz (0.8 - 14.5 meV) are carried out with the spectrometer (RT-20000, Tochigi Nikon Co. Ltd) which uses a standard technique for the transmis-

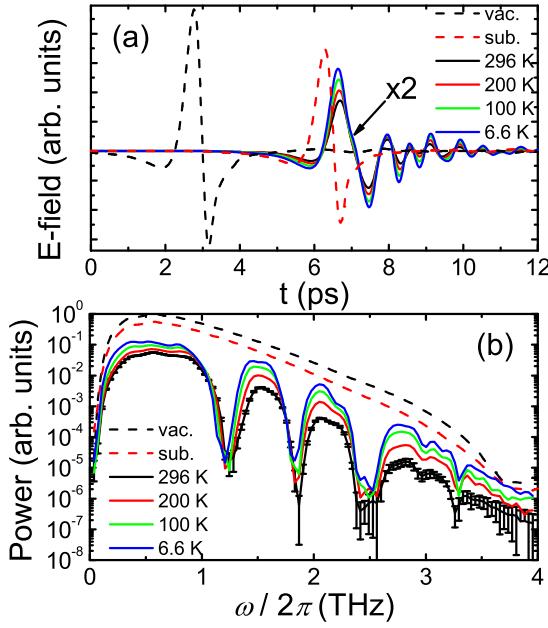


FIG. 1: (color online) (a) Time-evolution of THz-wave electric fields obtained from three different sets of measurements; vacuum, the substrate only and the type-I $\text{Ba}_8\text{Ga}_{16}\text{Ge}_{30}$ (BGG) sample glued onto the substrate and (b) the Fourier-transformed power spectra $|E(\omega)|^2$.

sion configuration, see, e.g., our preceding paper[9].

Figure 1(a) shows the time-evolution of transmitting electric fields through the vacuum, the substrate only and the sample glued onto the substrate, while Fig. 1(b) shows the Fourier-transformed power spectra. Then the refractive index spectra are determined by taking into account multiple reflections at the surface and boundary between the sample and substrate[9].

Figure 2 shows the conductivity spectra $\hat{\sigma}(\omega) = \sigma_1(\omega) + i\sigma_2(\omega)$. The real part $\sigma_1(\omega)$ takes several distinct peaks, while the imaginary part $\sigma_2(\omega)$ shows correspondingly the derivative with frequency. These discrete spectra, reminiscent of Lorentz spectra for optical phonons, are superimposed on the monotonous background conductivity. The magnitude of the background at room temperature is about $10 \Omega^{-1}\text{cm}^{-1}$ comparable to the dc conductivity and, with decreasing temperature, decreases almost down to zero, again consistent with the temperature dependence of the dc conductivity[8]. Therefore it is concluded that the conductivity spectra consist of both an almost constant Drude-like contribution due to the semiconducting carriers and the Lorentz-like spectra from optical phonon modes.

For the assignment of the phonon modes, we have made first-principle calculations[7] based on the density functional approximation for infrared-active optical modes of T_{1u} symmetry. Table I lists the phonon frequencies la-

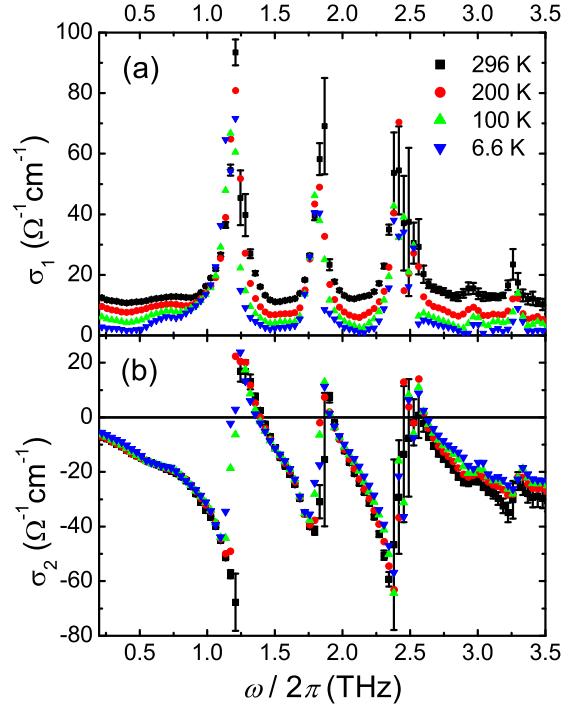


FIG. 2: (color online) Temperature-dependent conductivity spectra $\hat{\sigma}(\omega) = \sigma_1(\omega) + i\sigma_2(\omega)$ of BGG. For convenience, $1 \text{ THz} = 33 \text{ cm}^{-1} = 4.1 \text{ meV} = 48 \text{ K}$.

TABLE I: Calculated phonon frequencies and fitting parameters (6.6 K) of BGG. Due to the broad spectra around 2.4 THz (Fig. 2), the analysis for ν_3 and ν_4 is made with a single Lorentz curve (Eq. 2).

label	atom	calc. (THz)	$\omega_{0,i}$ (THz)	Γ_i (THz)	$S_i \times 10^{-13}$ ($\Omega^{-1}\text{cm}^{-1}s^{-1}$)
ν_1	Ba(2)	0.96	1.15	0.09	8.8
ν_2	Ba(2)	1.85	1.80	0.12	4.9
ν_3	cage	2.34	~ 2.4	0.16	~ 6
ν_4	Ba(1)	2.41			
ν_5	cage	2.70	2.98	0.06	0.4(1)
ν_6	cage	3.05	3.30	0.07	0.6(4)

beled as $\nu_i (i = 1 - 6)$ together with fitting parameters at $T = 6.6 \text{ K}$. The low-lying peaks observed at 1.15 and 1.80 THz can be assigned as the modes labeled ν_1 and ν_2 of the $\text{Ba}^{2+}(2)$ in the oversized cage vibrating within the plane perpendicular to the fourfold axis and along the fourfold axis, respectively. It is noted that the Einstein temperature of about 60 K (=1.2 THz) calculated from the atomic displacement parameters[6] is close to ν_1 . The broad peak around 2.4 THz can be considered as an overlapped spectrum consisting of both the collective cage mode ν_3 and the $\text{Ba}^{2+}(1)$ mode ν_4 in the smaller

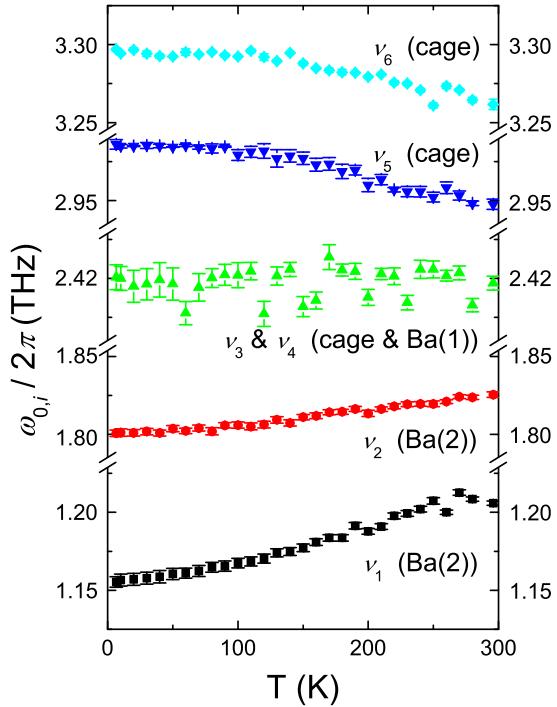


FIG. 3: (color online) Temperature dependence of peak frequencies of BGG for $\nu_i (i = 1 - 6)$.

cage. Higher frequencies of 2.98 and 3.30 THz are rather close to the collective cage modes ν_5 and ν_6 , respectively.

To extract each phonon contributions $\sigma_{1,i}(\omega)$ from the total spectra, we subtract the temperature-dependent Drude contributions by assuming a linear dispersion. Here we adjust the parameters of the linear curve in order to keep the sum rule that the spectral weight S_i of ν_i is conserved irrespective of temperature;

$$\int_0^\infty \sigma_{1,i}(\omega) d\omega = S_i = \pi N_i q_i^2 / 2M_i, \quad (1)$$

where M_i , N_i , and q_i are the mass, density and charge of atoms involved in the ν_i phonon mode. Then to the phonon spectra from ν_i we apply the Lorentz-type conductivity written by

$$\sigma_{1,i}(\omega) = \frac{(2/\pi) S_i \omega^2 \Gamma_i}{(\omega_{0,i}^2 - \omega^2)^2 + (\omega \Gamma_i)^2}, \quad (2)$$

where Γ_i is the relaxation rate and $\omega_{0,i}$ is the resonant frequency. Here S_i is analyzed from Eq.1 using the lowest-temperature data having negligibly small Drude contribution, see Table I. The observed ratio of $S_1/S_2 = 1.8$ is close to the theoretical value of 2 taken into account of the degrees of freedom for ν_1 and ν_2 modes. Furthermore the total contribution ($S_1 + S_2$) from these Ba(2)-ions' modes is about 13.7×10^{13} , which is larger by factor 4

than 3.4×10^{13} calculated with $M = M_{\text{Ba}}$, $N = 6/V_u$ ($V_u = 10.76^3 \text{ \AA}^3$ is the unit cell volume[6]) and $q = 2e$ for Ba(2) ions. Judging from the uncertainty in deriving the absolute value of σ_1 , it is not clear at present whether this discrepancy is intrinsic or not. To note, however, it may be an open issue to study how the charge and mass of the guest ion are effectively modified when coupled to the encapsulating cage. For ν_5 and ν_6 modes of cages, the small spectral weight as listed in the table may be reasonable.

The peak frequencies $\omega_{0,i}/2\pi$ are plotted as a function of temperature in Fig. 3. With decreasing temperature down to 6.6 K, the low-lying ν_1 and ν_2 modes soften by 4.2%, being consistent with the Raman-active modes at 1.05 THz (35 cm^{-1}) with E_g symmetry[7], and 1.8% respectively. On the other hand, other higher-frequency modes remain almost constant or harden slightly. In general the phonon hardening can be conventionally expected due to the lattice contraction at lower temperatures, so that the magnitude of the softening as observed in ν_1 and ν_2 might be underestimated due to an overcompensation for the hardening as observed in the cage modes ν_5 and ν_6 . Thus the present studies reproduce the softening phenomena in rattling modes of guest ions, which have been so far interpreted to be an evident proof for an anharmonicity effect in these localized modes.

In order to shed more light into the anharmonicity effect, we focus on the lowest-lying mode ν_1 to compare with model calculations for the optical conductivity. We assume the following Hamiltonian describing the motion of a single ion in a one-dimensional anharmonic potential (1D-AP) including the quadratic and quartic terms for the displacement x ,

$$H = \frac{p^2}{2M} + \frac{1}{2}kx^2 + \frac{1}{4}\lambda x^4, \quad (3)$$

where p and M are the momentum and mass (here, suppose Ba ions). The constants k and λ are assumed to be positive for a single well. Using a linear response theory, the optical conductivity $\hat{\sigma}(\omega)$ from the anharmonic phonon is expressed as

$$\hat{\sigma}(\omega)/\sigma_0 = i\omega \sum_{\omega_{nm} > 0} |\langle n|x/x_0|m \rangle|^2 \frac{e^{-\beta E_m} - e^{-\beta E_n}}{Z} \times ((\omega - \omega_{nm} + i\Gamma_0/2)^{-1} - (\omega + \omega_{nm} + i\Gamma_0/2)^{-1}), \quad (4)$$

where $|n\rangle$ and E_n are the eigenstate and eigenvalue of the Hamiltonian, $\beta = 1/k_B T$, $Z = \sum_n e^{-\beta E_n}$ and $\omega_{nm} = (E_n - E_m)/\hbar$. Here a phenomenological relaxation rate Γ_0 is introduced corresponding to the line width of the spectral shape. The conductivity is normalized with $\sigma_0 = q^2 N x_0^2 / \hbar$, where x_0 is a length scale, $x_0^2 = \hbar / \sqrt{Mk}$. The calculation indicates that the transition probability arises mostly from the matrix elements $\langle n+1|x|n \rangle$ between neighboring levels. Details are discussed in Ref. [10].

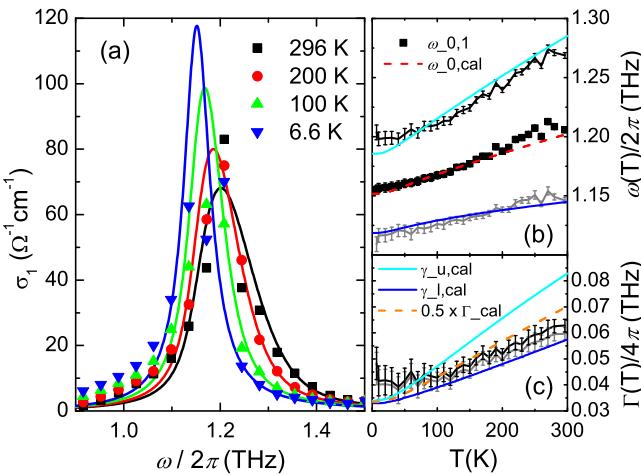


FIG. 4: (color online) (a) Spectral shapes of $\sigma_1(\omega)$ from ν_1 of BGG. The filled symbols and solid lines indicate the data and calculations respectively. Temperature dependence of (b) the peak frequency ω_0 and (c) the observed line widths compared with calculated ones. See the text for the definitions.

For the numerical calculation of $\sigma_1(\omega, T)$ for the ν_1 mode, we fix the parameters: $(1/2\pi)\sqrt{k/M} = 1.143$ THz, $\lambda/2k \times a_B^2 = 0.21$ (a_B is Bohr radius), and the temperature-independent constant $\Gamma_0/2\pi = 0.067$ THz. With these parameters the observed spectra are best fitted to the calculations as shown in Fig. 4(a). Figure 4(b) shows the temperature dependences of the peak position ($\omega_{0,cal}$) and the upper (lower) frequencies $\omega_{u(l),cal}$ giving half the peak height of the calculated spectra, together with corresponding data $\omega_{0,1}$, $\omega_{u,1}$ and $\omega_{l,1}$ obtained from the Lorentzian fitting. The data can be reproduced with our calculations.

In a usual harmonic approximation with $\lambda = 0$, $E_n = \hbar\omega_E(n + 1/2)$ and all the neighboring level-spacings become equivalent resulting in the well-known Lorentz-type dispersion in the complex conductivity for an harmonic oscillator $i\omega/(\omega^2 - \omega_E^2 + i\omega\Gamma_0)$, as Eq.2. In contrast, the anharmonicity due to the quartic term in Eq. (3) induces unequally spaced energy levels, resulting in n -dependent excitations that the larger the n , the wider the spacing. At enough high temperatures in comparison to $\hbar\omega_E/k_B = 55$ K (= 1.15 THz), phonons are thermally distributed widely from $n = 0$ to higher levels in accordance to the Boltzmann factor, while, at lower temperatures, phonons become condensed mostly around $n = 0$, and can be effectively excited to $n = 1$. The overall spectral shapes, given by a superposition of each Lorentzian curves having the n -dependent peak frequencies, become broad at higher temperature. Thus the anharmonicity effects on the conductivity spectra are expected both in the shift of the peak frequency and in the change of the line width.

Thus the 1D-AP model predicts an asymmetric change in the line width with temperature, which can be featured in the temperature dependence of $\gamma_{u(l),cal} =$

$|\omega_{u(l),cal} - \omega_{0,cal}|$ measuring the width in the higher (lower)-frequency region with respect to the peak frequency. Figure 4(c) shows stronger temperature dependence in $\gamma_{u,cal}$ than in $\gamma_{l,cal}$, indicating the asymmetric change of the spectra. The data are well reproduced by the mean width defined as half the $\Gamma_{cal} = \gamma_{u,cal} + \gamma_{l,cal}$ shown by the dashed line.

Finally, we remark upon two assumptions made in the above discussions. First, we have assumed the constant line-width (Γ_0) in conductivity calculations. This damping effect may be attributed, in part at least, to some mode-mode couplings to the anharmonic phonons themselves and also to the acoustic cage phonons. Second, we have assumed the positive quadratic potential ($k > 0$ in Eq. 3) meaning the on-center rattling in a single minimum potential. For the case of the off-centering with two-level minima with $k < 0$, the peak frequency and the line width are hardly fitted to the data.

In conclusion, the temperature dependences of the peak frequency and the line width in the conductivity spectra of BGG are explained by 1D-AP model calculations without any interactions taken into account. Importantly to note, these dependences are determined solely by the Bose-factor for optical excitations of anharmonic phonons with the nonequally spaced energy levels.

The works have been supported by Grants-in-Aid for Scientific Research (A)(15201019,1820432), the priority areas (1951011, 15072205, 20102004) from MEXT, Japan, the Sasakawa Scientific Research from Japan Science Society, and also by the Global COE program “Materials Integrations”, Tohoku University.

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